

Thickness Dependent Gate Oxide Quality of Thin Thermal Oxide Grown on High Temperature Formed SiGe

Y. H. Wu, Albert Chin, *Senior Member, IEEE*, and W. J. Chen

Abstract—For thin oxides grown on high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$, the gate oxide quality is strongly dependent on oxide thickness and improves as thickness reduces from 50 to 30 Å. The thinner 30 Å oxide has excellent quality as evidenced by the comparable leakage current, breakdown voltage, interface-trap density and charge-to-breakdown with conventional thermal oxide grown on Si. The achieved good oxide quality is due to the high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$ that is strain relaxed and stable during oxidation. The possible reason for strong thickness dependence may be due to the lower GeO_2 content formed in thinner 30 Å oxide rather than strain relaxation related rough surface or defects.

Index Terms—Gate oxide integrity, SiGe oxide, oxide reliability.

I. INTRODUCTION

HIGH quality thin gate oxide [1]–[3] is urgently required for SiGe p-MOSFET's [4]–[10], although higher hole mobility and better current drive capability than standard Si counterpart are demonstrated. To avoid strain relaxation in SiGe, gate oxide is generally formed by a low temperature process [11] using plasma CVD deposition, but the quality is unable to compare with conventional thermal SiO_2 . On the other hand, high temperature oxide directly grown on SiGe has poor quality because the strain relaxation not only reduces hole mobility [10] but also generates rough surface and defects [8]–[10]. The low temperature process also prohibits SiGe from using in deep sub- μm devices with high-K dielectrics [12], [13]. Recently, we developed a high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$ [14] using the similar process as silicidation [15]. Because SiGe is formed by high temperature rapid-thermal-annealing (RTA) and solid-phase epitaxy (SPE), better thermal stability can be expected. Therefore, high temperature (950 °C) RTA can be used to achieve the important low junction-leakage [11] of p^+n source and drain after implantation [16]. Because of the high Ge content, good hole mobility of 250 cm^2/Vs is still obtained [16] that is two times higher than Si even without the strain [17]. In this work, we have studied thin gate oxides grown on high temperature formed SiGe. The gate oxide quality improves and becomes comparable with standard SiO_2 as thickness reduces from 50 to 30 Å. Possible reason for this

improvement may be due to the lower GeO_2 content inside the thinner 30 Å oxide.

II. EXPERIMENTAL

To avoid any B diffusion though thin gate oxide, p-type Si wafers were used instead of n-type wafers although SiGe should be applied to p-MOSFET's. After device isolation, $\text{Si}_{0.3}\text{Ge}_{0.7}$ was formed in the active region by RTA and SPE at 900 °C from deposited amorphous Ge layer. The native oxide is specially taken care of using HF-vapor passivation [13]–[16] before Ge deposition. More detailed $\text{Si}_{0.3}\text{Ge}_{0.7}$ forming process and material characterization can be found elsewhere [14]. Gate oxides of 30 and 50 Å were grown by dry oxygen at 900 °C for both $\text{Si}_{0.3}\text{Ge}_{0.7}$ and Si control sample. The oxide thickness is measured by ellipsometer and TEM. Gate capacitors were formed after a 3000 Å poly-Si deposition and subsequent process steps.

III. RESULTS AND DISCUSSION

Fig. 1(a) and (b) show the current–voltage (I – V) characteristics of 50 and 30 Å thermal oxides formed on both $\text{Si}_{0.3}\text{Ge}_{0.7}$ and standard Si, respectively. For control 50 and 30 Å SiO_2 , a breakdown electric field of ~ 14.5 MV/cm is obtained that is the typical value for thermal SiO_2 on Si. For 50 Å oxide grown on $\text{Si}_{0.3}\text{Ge}_{0.7}$, three to four times higher leakage current and 1 V lower breakdown voltage than control SiO_2 are observed. In sharp contrast, comparable leakage current and breakdown voltage with control SiO_2 are obtained for ~ 30 Å thermal oxide grown on $\text{Si}_{0.3}\text{Ge}_{0.7}$. The smaller leakage current of thermal oxide grown on $\text{Si}_{0.3}\text{Ge}_{0.7}$ than Si is due to the 1 Å thicker thickness. The trend of improved oxide quality with thinner thickness is important for deep sub- μm devices where the gate oxide thickness is continuously scaling down.

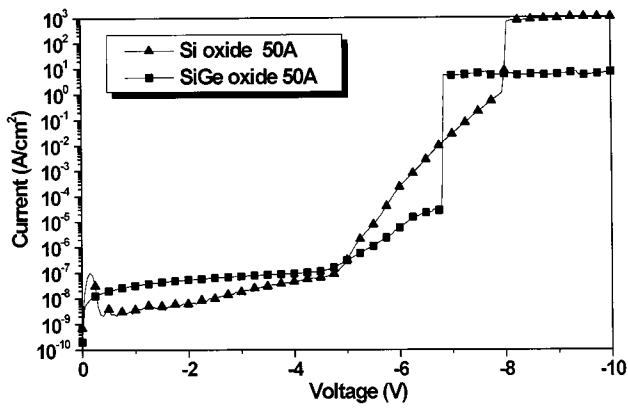
We have used capacitance–voltage C – V and TEM to further investigate the oxide quality. Important interface-trap density (D_{it}) can be obtained from high and low frequency C – V curves. Fig. 2 shows the D_{it} and cross-sectional TEM of SiGe thermal oxides. Although very low D_{it} of 4×10^{10} and 7×10^{10} $\text{eV}^{-1}/\text{cm}^2$ is measured for respective 30 Å and 50 Å oxides, this D_{it} difference is small. The low D_{it} may be due to very smooth interface and uniform oxide thickness observed by TEM. Therefore, the strong thickness dependence on oxide quality is irrelevant to strain relaxation related rough interface or defect generation. This is due to the high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$ that is already strain relaxed and stable during ox-

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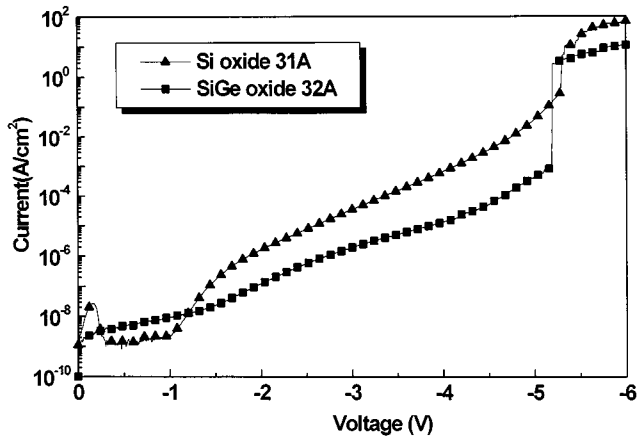
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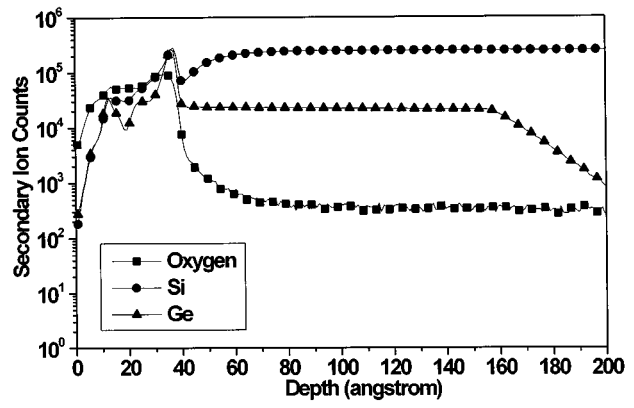


(a)

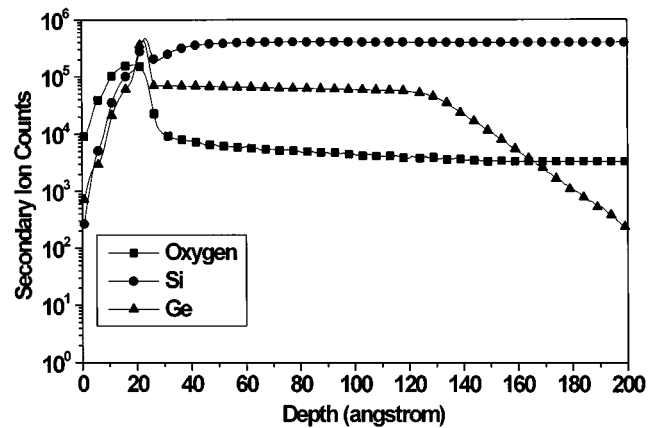


(b)

Fig. 1. I - V characteristics of (a) 50 and (b) 30 Å thermal oxides formed on both $\text{Si}_{0.3}\text{Ge}_{0.7}$ and standard Si.



(a)



(b)

Fig. 3. SIMS profiles of (a) 50 and (b) 30 Å thermal oxides grown on high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$.

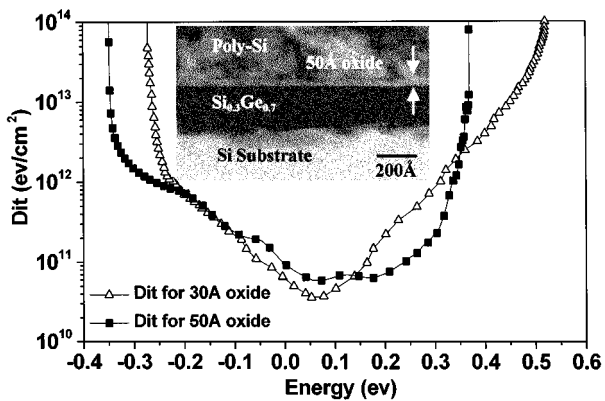


Fig. 2. Interface trap density for 30 and 50 Å oxide grown on high temperature formed $\text{Si}_{0.3}\text{Ge}_{0.7}$. Inset figure is the cross-sectional TEM of 50 Å SiGe thermal oxide on $\text{Si}_{0.3}\text{Ge}_{0.7}$.

dation. Furthermore, no dislocation penetration into underneath Si is observed that also explains the low leakage current.

We have further analyzed the SiGe oxides by SIMS. Fig. 3(a) and (b) shows the SIMS profiles of 50 and 30 Å SiGe oxides, respectively. In both cases, a Ge snow plowing is observed in oxide-SiGe interface. For 50 Å SiGe oxide, in addition to the Ge pileup at interface, another Ge peak at ~ 15 Å below surface is also found. Similar phenomenon is also reported for oxide

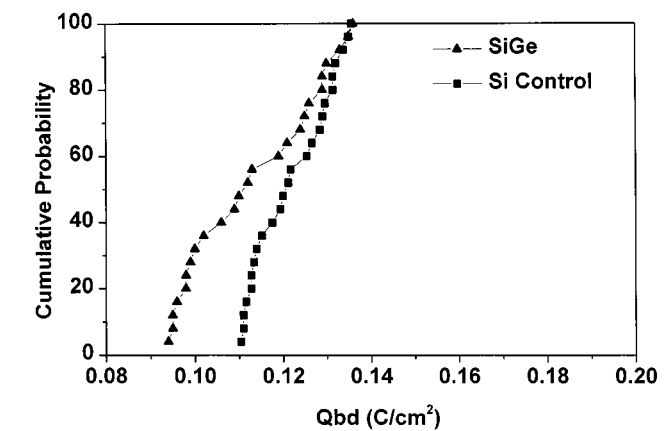


Fig. 4. Charge-to-breakdown (Q_{BD}) distribution of thermal oxides grown on $\text{Si}_{0.3}\text{Ge}_{0.7}$ and Si after a 4.5 V stress.

grown on $\text{Si}_{0.5}\text{Ge}_{0.5}$ [18]. In contrast, no such additional peak can be observed for thinner 30 Å SiGe oxide. The possibility of additional Ge peak less than 15 Å below surface is low, although this small distance is near the limitation of SIMS. At such small distance close to surface, Ge may have high probability of out-diffusion because of the high oxidation temperature. Therefore, the improved oxide quality of 30 Å SiGe oxide may be due to the smaller content of weaker GeO_2 inside the SiO_2 .

To further evaluate the 30 Å SiGe oxide, we have also investigated the oxide reliability. Fig. 4 shows the charge-to-breakdown (Q_{BD}) distribution for both thermal SiGe oxide and SiO₂ under -4.5 V stress. An average Q_{BD} of 0.12 C/cm² is obtained for control thermal SiO₂ that is close to reported value. Good SiGe oxide reliability can be evidenced by the high Q_{BD} of 0.11 C/cm² that is comparable with thermal SiO₂. The good reliability is related to the high temperature stable Si_{0.3}Ge_{0.7} that has a uniform electric field distribution resulted from the smooth surface and interface.

IV. CONCLUSION

Improved oxide quality is observed with decreasing thickness of thermal oxide grown on high temperature formed Si_{0.3}Ge_{0.7}. We have achieved comparable oxide quality with thermal SiO₂ for thinner 30 Å oxide.

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