

To enlarge the lateral grain growth length we introduce a heat reservoir layer to increase the grain growth time. Figure 3.12 is the fabrication procedure. The steps are similar to figure 3.9 and the only difference is that a 3500Å capping SiO₂ layer was deposited on the α-Si film before the laser irradiation and the heat reservoir layer was removed by the diluted HF in 15 minutes after laser irradiation.

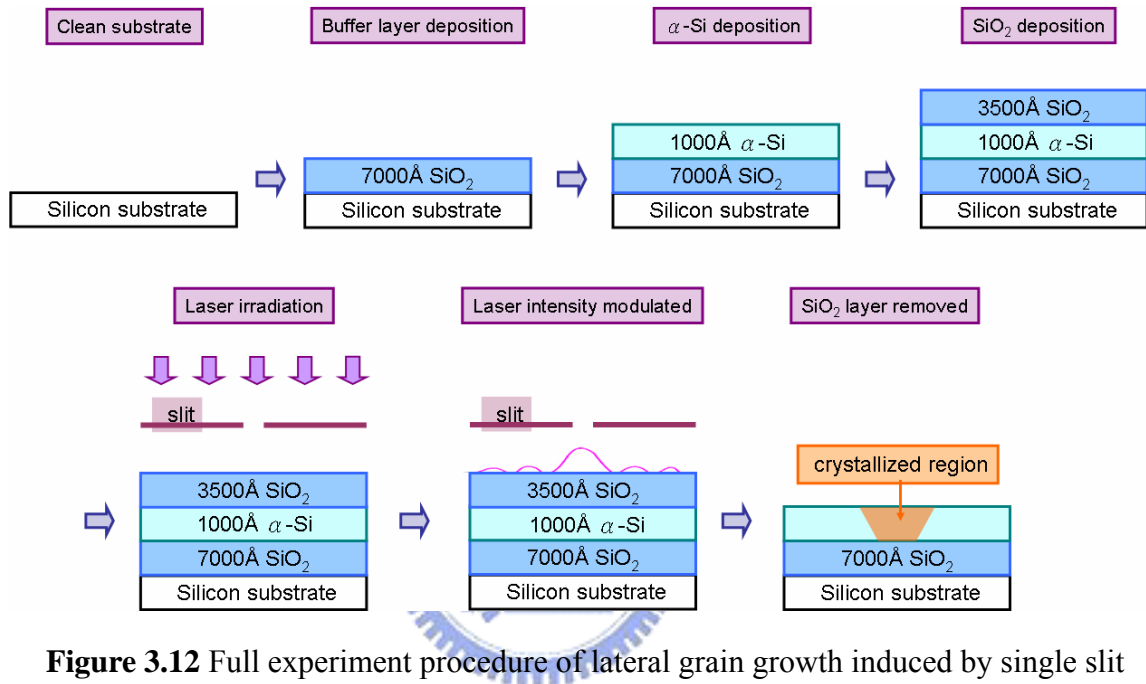


Figure 3.12 Full experiment procedure of lateral grain growth induced by single slit diffraction with capping oxide layer (including the sample preparation and laser irradiation).

The SEM images of the single slit diffraction with the heat reservoir layer are shown in figure 3.13(a) and (b). The reservoir layer could absorb the exceeding thermal energy transferred from the Si film. When the temperature of the Si film became lower than that of SiO₂ the thermal energy stored in the capping layer would release and transfer back to the Si layer. The film thickness of the heat reservoir layer is a critical parameter. The thinner film couldn't store enough thermal energy such that the effect of the reservoir layer would be limited. The thicker SiO₂ film wouldn't be a heat reservoir layer any more but the heat sink due to great amount of thermal energy transferred to the capping oxide. The heat transferred to the SiO₂ layer would increase the temperature of SiO₂ but the temperature was not high enough to heat the Si layer again as the temperature of the Si layer decreased. As our

expectation the lateral grain size increases to $2\mu\text{m}$. It's 1.5 times that without the capping SiO_2 . It's similar to the result which was presented by Masakiyo Matsumura in 2002[16] shown in figure 3.14. The lateral grain growth length increased from $4\mu\text{m}$ to $7\mu\text{m}$.

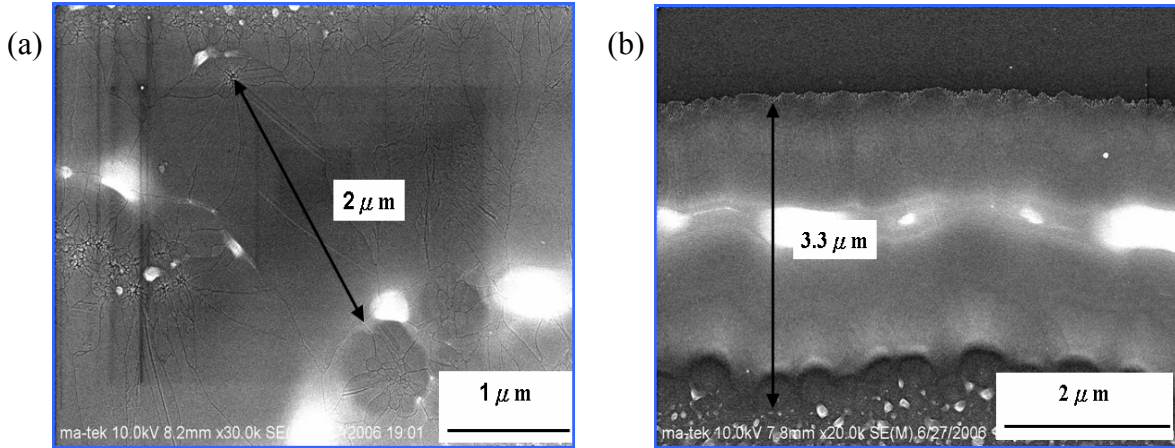


Figure 3.13 The SEM images for the lateral grain growth region with a capping SiO_2 layer. (a) $L = 4.124\text{mm}$, the lateral grain growth length = $2\mu\text{m}$. (b) The lateral grain growth length = $3\mu\text{m}$ with a single grain boundary in it.

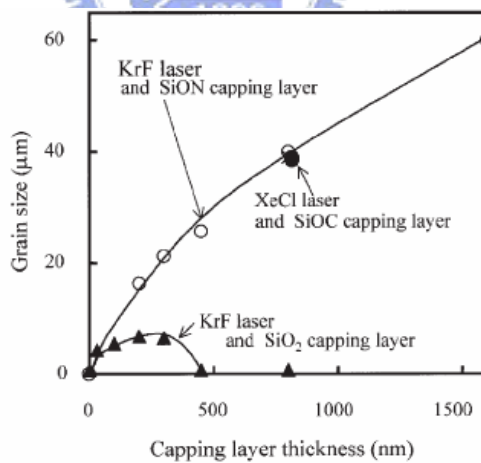


Figure 3.14 The effect of capping layer in PMELC. The initial grain size is $4.5\mu\text{m}$ without capping layer. The optimum SiO_2 film thickness is 370nm and the grain size is $7\mu\text{m}$ which is about 1.56 times the former.

In the lateral crystallization induced by the single slit diffraction, the lateral crystallization position and the grain length could be principally controlled by the laser energy

and laser intensity profile which depends on the “L/a” separately. Furthermore, the grain size could be enlarged by depositing the heat reservoir layer SiO₂ which could reserve the exceeding heat stored in it during the crystallization process. In figure 3.15, we compare the heat flux at 25ns (just after irradiation) and 200ns. At 25ns the laser energy was only absorbed by the Si layer and the heat flux indicates that the thermal energy tends to transfer to the SiO₂ layer. At time = 140ns, the temperature of the Si layer was lower than that of the capping layer such that the heat stored in the capping SiO₂ transfer to the Si layer to increasing the melting time of the Si layer. The longer melting time would increase the grain growth time.

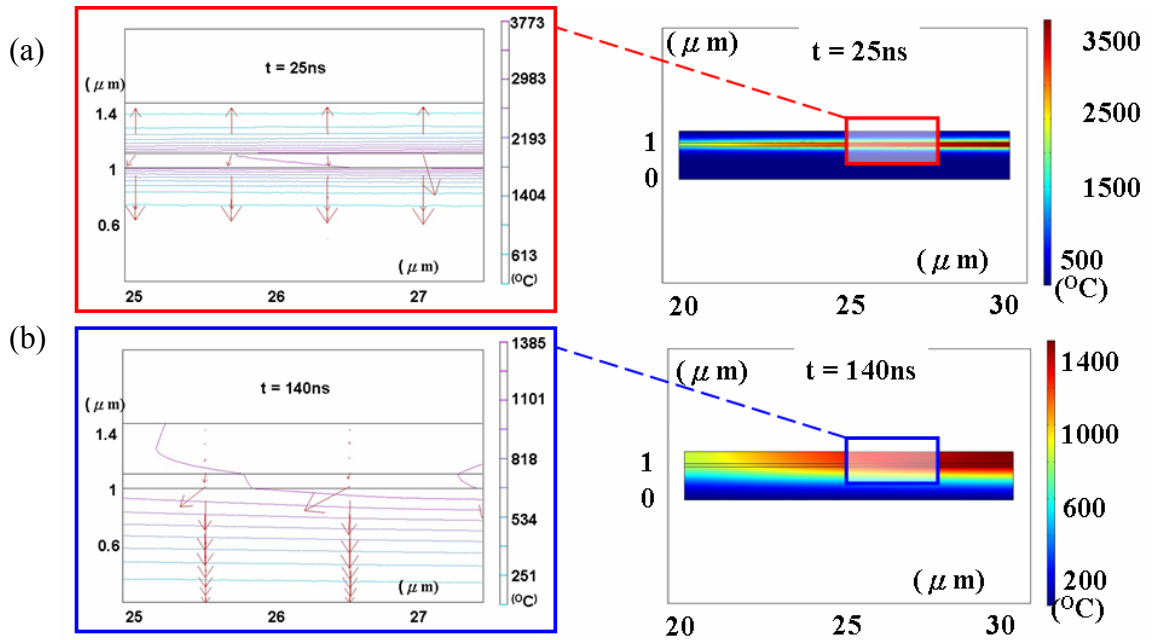


Figure 3.15 Heat flux at (a) time = 25ns and (b) time = 140ns.

When the α -Si film thickness increases to 2000Å the lateral grain growth would be enlarged. The curves of lateral grain growth length versus L with and without the capping reservoir layer is shown in Figure 3.16. We found that the lateral grain length of the poly Si with the capping SiO₂ is 1.2 times that without capping layer because the 3500Å SiO₂ might not be enough for the 2000Å Si thickness. If we would like to enlarge the grain size more, a thicker SiO₂ film is necessary.

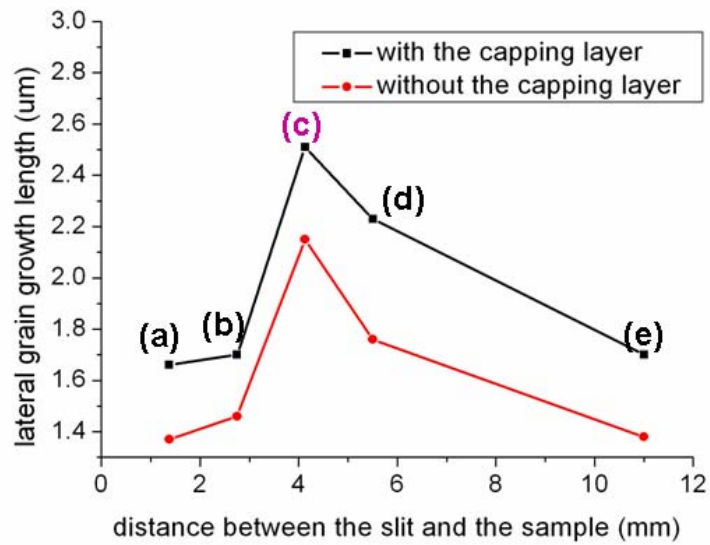
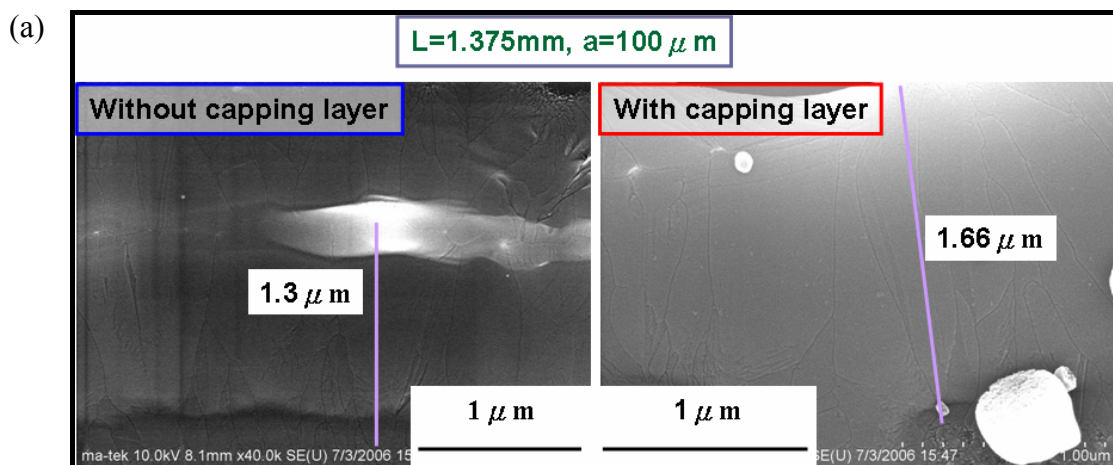
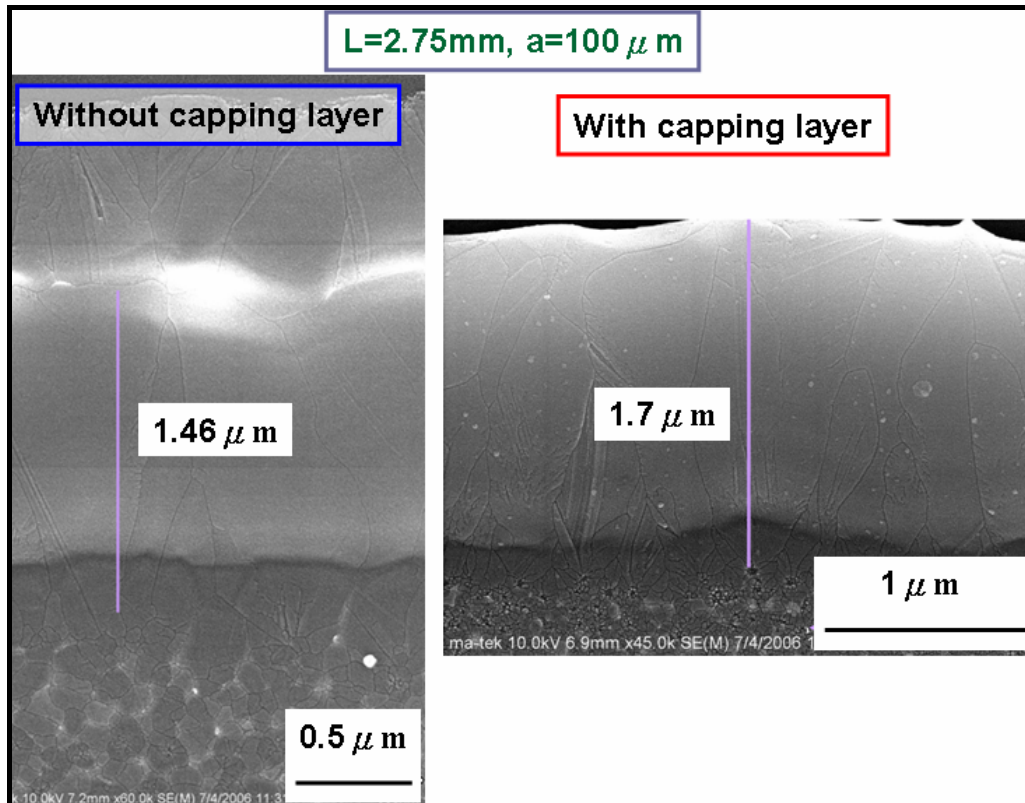


Figure 3.16 Lateral grain size versus the distance between slit and sample with and without the capping layer.

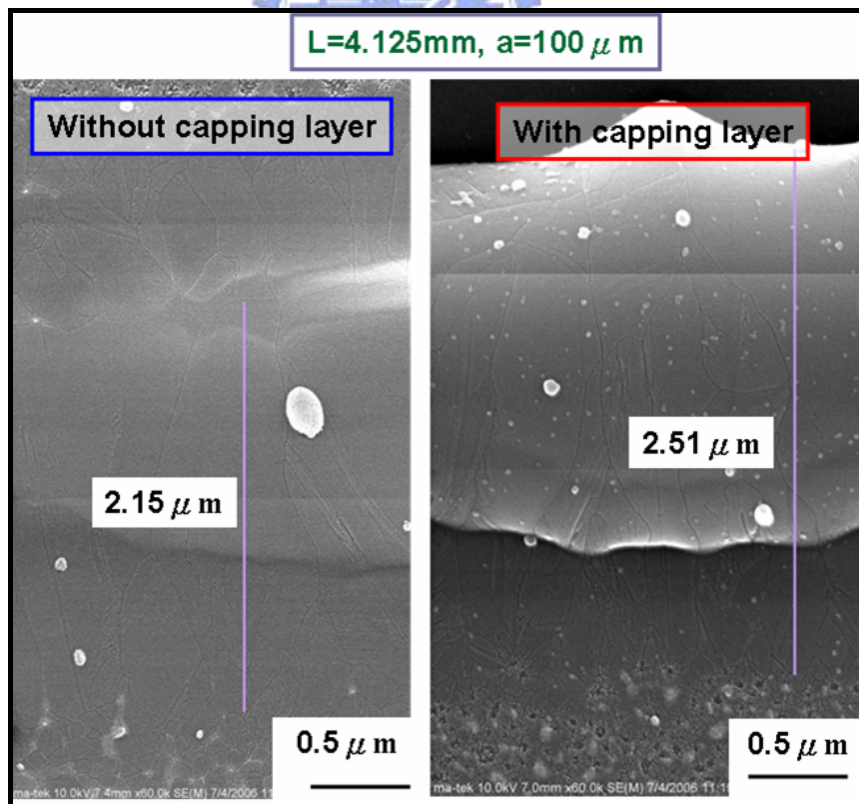
The fine view of the SEM images for the lateral grain growth region is shown in figure 3.17, the laser energy is about $650\text{mJ}/\text{cm}^2$. L and a are labeled on it. The left figures are the sample without the reservoir film, and the right ones are that with the capping SiO_2 .



(b)



(c)



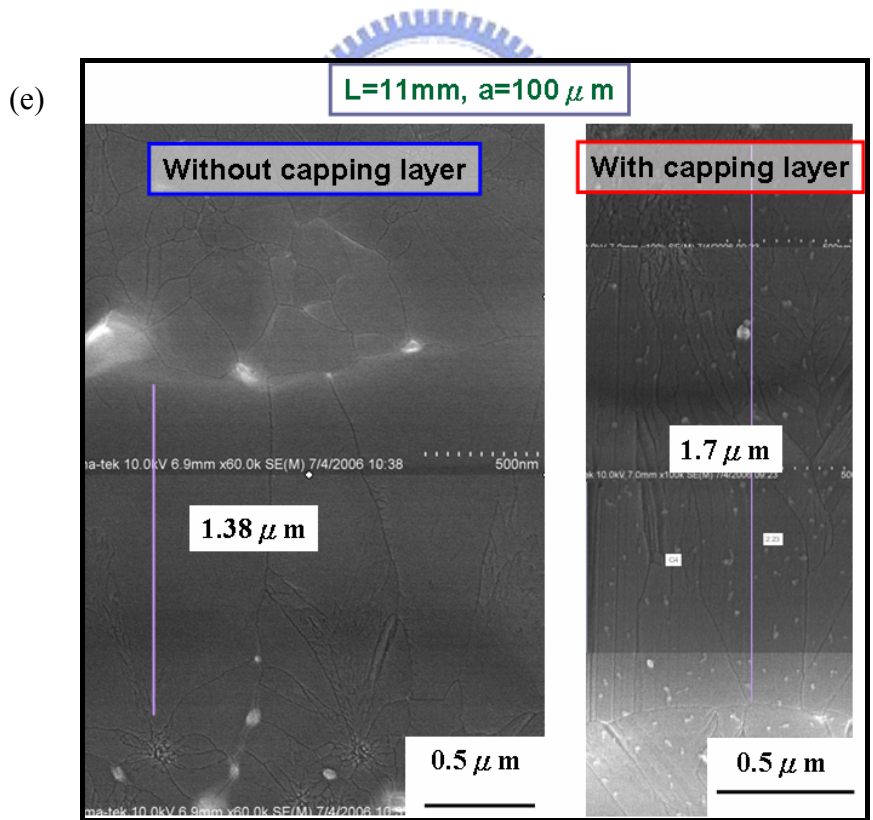
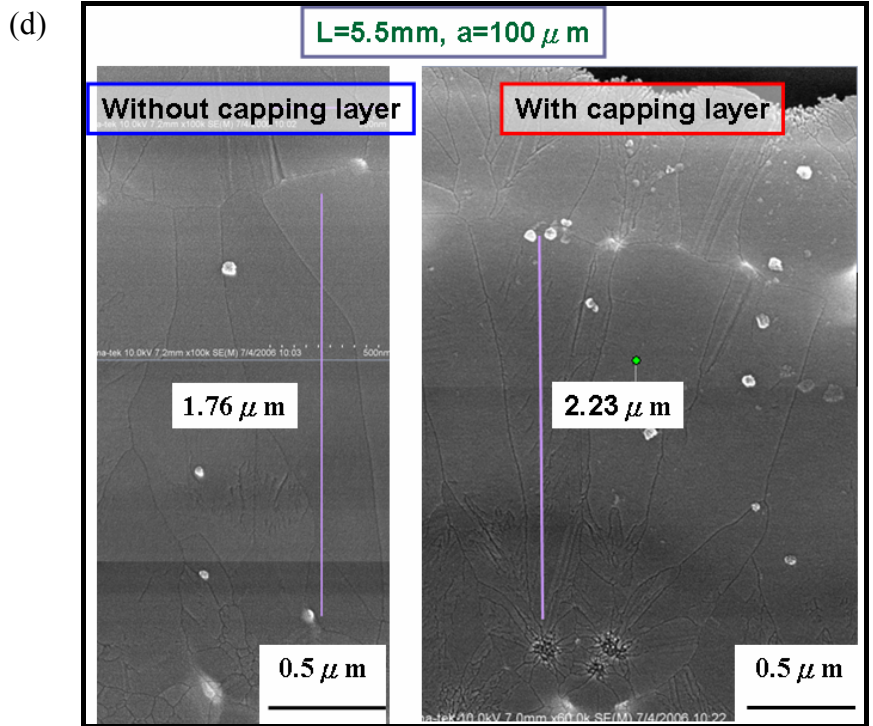


Figure 3.17 The fine view of the SEM image for the lateral grain growth region. The α -Si film thickness = 2000\AA , laser intensity = $650\text{mJ}/\text{cm}^2$, $a = 100\ \mu\text{m}$, $L =$ (a) 1.375mm , (b) 2.7mm , (c) 4.125mm , (d) 5.5mm , (e) 11mm respectively.

Chapter 4 Conclusion and future work

In our simulation, we didn't consider the condition that the value of thermal conductivity and the heat specific would change with the temperature. The simulation would be completed until this term is combined in it. In the metal reflection method, the grain growth length and grain growth direction could be controlled well. We found that lateral grain length increasing with the laser energy density increasing as shown in figure 3.4 (f). However, the ablation of metal Cr film would occur at the laser intensity larger than $300\text{mJ}/\text{cm}^2$. There are fine grains existing in region near the edge of metal patterns due to the random nucleation induced by the rapid temperature decreasing. Nuclei couldn't be controlled at the expected position precisely. For the nano-hole structure, the average grain size modulated by the value of L and a , the grain size is equal to $(L-a)/2$. The nucleation position was controlled at the edge of holes. In single slit diffraction method, the lateral crystallization mainly depends on the laser intensity distribution slope. In figure 3.11(f), when the slope is not large enough the lateral grain growth phenomenon would not be obvious. With a reservoir 3500\AA SiO_2 film, the average grain size increase to $2\mu\text{m}$ which is about 1.5 times that without capping SiO_2 layer.

Compare the three method we tried to approach the lateral grain growth in table 4.1. We could find that the largest lateral grain length is induced by the single slit methods, the lateral growth direction control is induced by the metal reflection layer and the nuclei position control is induced by the nano-hole structure. For metal reflection layer, a new metal material which could endure the higher laser energy is necessary to enlarge the lateral grain growth size. For the nano-structure, a proper value of L (the distance between each hole) and D (the size of the nano-holes) could approach a largest grain size as possible. For the single slit diffraction method, the subsequence and larger lateral grains could be obtained by tuning the laser intensity distribution and the laser energy and the effect of the heat reservoir layer could also enhanced by change the film thickness and the material. The distance between the slit

and the sample should be controlled with a movable stage such that the intensity distribution could be more easily and precisely. Furthermore, the design of the slit size and shape would be useful way to approach 2D grain growth control. Combining the laser intensity modulation and the structure method is a useful way to create a single grain in the TFT channel and these methods would take into the device fabrication in future.

Table 4.1 Comparison of three crystallization methods.

| | Average lateral grain length | Nuclei position control | Grain growth direction control |
|--|---------------------------------------|-------------------------|--------------------------------|
| Metal reflection layer | $0.7 \mu\text{m}$ | normal | great |
| Nano-hole structure | $0.4 \mu\text{m}$ | great | good |
| Single-slit diffraction (with capping layer) | $1.3 \mu\text{m}$ ($2 \mu\text{m}$) | normal | good |

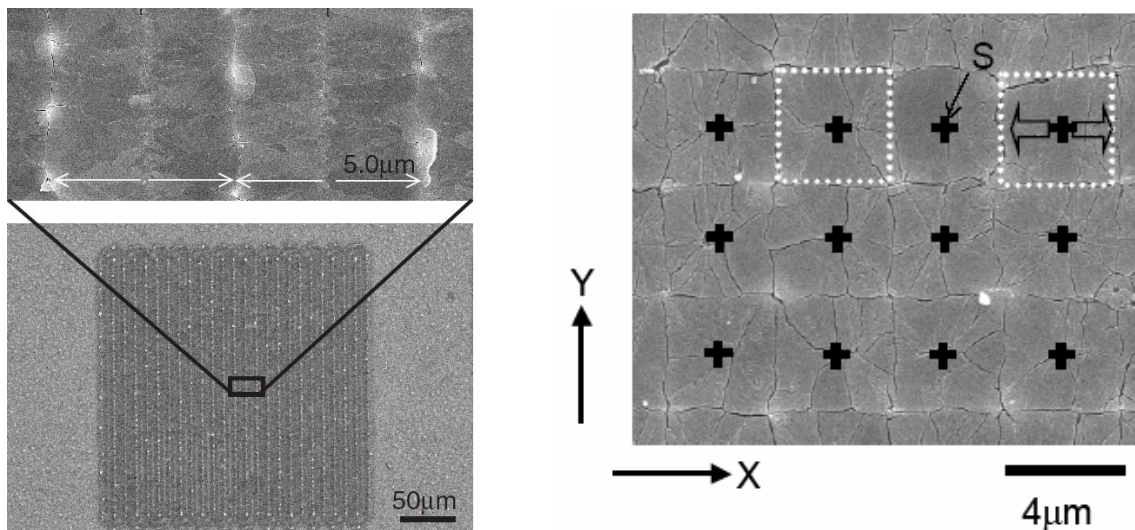


Figure 4.1 The subsequent lateral grain growth region induced by tuning the laser intensity well[26] and 2D grain growth control.

References

- [1] A.T. Voutsas, M.K. Hatalis, "Structure of As-Deposited LPCVD Silicon Films at Low Deposition Temperatures and Pressures". *J. Electrochem, Soc.* 139 (1992) 2659.
- [2] A.T. Voutsas, M.K. Hatalis, "DEPOSITION AND CRYSTALLIZATION OF A-SI LOW-PRESSURE CHEMICALLY VAPOR-DEPOSITED FILMS OBTAINED BY LOW-TEMPERATURE PYROLYSIS OF DISILANE", *J. Electrochem, Soc.* 140 (1993) 871.
- [3] A.T. Voutsas, M.K. Hatalis, "STRUCTURAL CHARACTERISTICS OF AS-DEPOSITED AND CRYSTALLIZED MIXED-PHASE SILICON FILMS", *J. Electrochem, Soc.* 23 (1994) 319.
- [4] Byoung Dong Kim, "Solid phase crystallization of amorphous silicon on glass by thin film heater for thin film transistor (TFT) application", Hunjoon Jung, Gi-Bum Kim, Seung-Ki Joo, *Microelectronics Journal*, 34 (2003) 767.
- [5] Miyasaka M, Shimoda T, Makihiro K, Asano T, Pecz B, Stoemenos, "Structural Properties of Nickel Metal-Induced Laterally Crystallized Silicon Films and Their Improvement Using Excimer Laser Annealing", *J. Jpn. J. Appl. Phys. Vol. 42* (2003) 2592
- [6] Naoyuki Higashi, Gou Nakagawa, Tanemasa Asano, Mitsutoshi Mitasaka, John Stoemenos, "Location and Orientation Control of Si Grain by Combining Metal-Induced Lateral Crystallization and Excimer Laser Annealing", *Jpn. J. Appl. Phys. Vol. 45*(2006) 4347.
- [7] Akito Hara, Fumiya Takeuchi, Michiko Takei, Katsuyuki Suga, Kenichi Yoshino, Mitsuru Chida, Yasuyuki Sano, Nobuo Sasaki, "High-Performance Polycrystalline Silicon Thin Film Transistors on Non-Alkali Glass Produced Using Continuous Wave Laser Lateral Crystallization", *Jpn. J. Appl. Phys. Vol. 41* (2002) L311.

- [8] Akito Hara, Kenichi Yoshino, Fumiyo Takeuchi, Nobuo Sasaki, “Selective Single-Crystalline-Silicon Growth at the Pre-defined Active Region of a Thin Film Transistor on Glass by Using Continuous Wave Laser Irradiation”, Jpn. J. Appl. Phys. Vol. 42 (2003) 23.
- [9] Ryoichi Ishihara, Yasushi Hiroshima, Daisuke Abe, Barry D. van Dijk, Paul C. van der Wilt, Seiichiro Higashi, Satoshi Inoue, Tatsuya Shimoda, J. W. Metselaar, C. I. M. Beenakker, “Single-grain Si TFTs with ECR-PECVD gate SiO₂”, IEEE Trans. Electron Devices 51 (2004) 500.
- [10] Ming He, Ryoichi Ishihara, Yasushi Hiroshima, Satoshi Inoue, Tatsuya Shimoda, Wim Metselaar, Kees Beenakker, “Effects of Capping Layer on Grain Growth with μ -Czochralski Process during Excimer Laser Crystallization”, Jpn. J. Appl. Phys. Vol. 45 (2003) 1.
- [11] Do-Hyun Choi, Kazuhiro Shimizu, Osamu Sugiura, Masakiyo Matsumura, “Drastic Enlargement of Grain Size of Excimer-Laser-Crystallized Polysilicon Films”, Jpn. J. Appl. Phys. Vol. 31 (1992) 4545.
- [12] Akito Hara, Nobuo Sasaki, “Control of Nucleation and Solidification Direction of Polycrystalline Silicon by Excimer Laser Irradiation”, Jpn. J. Appl. Phys. Vol. 39 (2000) L1
- [13] Chang-Ho Oh, Motohiro Ozawa, Masakiyo Matsumura, “A Novel Phase-Modulated Excimer-Laser Crystallization Method of Silicon Thin Films”, Jpn. J. Appl. Phys. Vol. 37 (1998) L492
- [14] Chang-Ho Oh, Mitsuru Nakata, Masakiyo Matsumura, “Optimization of phase-modulated excimer-laser annealing method for growing highly-packed large-grains in Si thin-films”, Applied Surface Science 154-155 (2000) 105-111.
- [15] Wen-Chang Yeh, Masakiyo Matsumura, “Effects of a Low-Melting-Point Underlayer on Excimer-Laser-Induced Lateral Crystallization of Si Thin-Films”. Jpn. J. Appl. Phys. Vol. 40 (2001) 3096

- [16] WenChang Yeh, Masakiyo Matsumura, "Proposed Sample Structure for Marked Enlargement of Excimer-Laser-Induced Lateral Grain Growth in Si Thin Films", Jpn. J. Appl. Phys. Vol. 41 (2002) 1909
- [17] Chang-Ho Oh, Masakiyo Matsumura, "A proposed single grain-boundary thin-film transistor", IEEE Electron Device Lett. Vol 22 (2001) 20.
- [18] Satoshi Yoshimoto, Chang-Ho Oh, Masakiyo Matsumura, "A New Sample Structure for Position-Controlled Giant-Grain Growth of Silicon using Phase-Modulated Excimer-Laser Annealing", Jpn. J. Appl. Phys. Vol. 40 (2001) 4466.
- [19] Motohiro Ozawa, Chang-Ho Oh, Masakiyo Matsumura, "Two-Dimensionally Position-Controlled Excimer-Laser-Crystallization of Silicon Thin Films on Glassy Substrate", Jpn. J. Appl. Phys. Vol. 38 (1999) 5700.
- [20] Robert S, Sposili, James S, Im, "Sequential lateral solidification of thin silicon films on SiO₂", Appl. Phys. Lett. 69 (1996) 2864.
- [21] R Dassow, J R Kohler, Y Helen, K Mourgues, O Binnaud, T Mohammed-Brahim, J H Wener, "Laser crystallization of silicon for high-performance thin-film transistors", Seimicond. Sci. Technolo. 15 (2000) L31.
- [22] Jae-Hong Jeon, Min-Cheol Lee, Kee-Chan Park, Min-Koo Ham, "A new polycrystalline silicon TFT with a single grain boundary in the channel", IEEE Electron Device Letters Vol. 22 (2001) 429.
- [23] Wen-Chang Yeh, Masakiyo Matsumura. "Numerical Calculation of Excimer-Laser-Induced Lateral-Crystallization of Silicon Thin-Films", Jpn. J. Appl. Phys. Vol. 40 (2001) 492.
- [24] Ching-Wei Lin, Li-Jing Chen, Yin-Lung Lu, Yih-Shing Lee, Huang-Chun Chen, "High-performance low-temperature poly-Si TFTs crystallized by excimer laser irradiation with recessed-channel structure", IEEE Electron Device Letters Vol. 22 (2001) 269.
- [25] Eugene Hecht, "Optics 4th edition", (2002) p.447.

- [26] Masayuki Jyumonji, Yoshinobu Kimura, Yukio Taniguchi, Masato Hiramatsu, Hiroyuki Ogawa, Masakiyo Matsumura. “Optimum Light Intensity Distribution for Growing Large Si Grains by Phase-Modulated Excimer-Laser Annealing”, Jpn. J. Appl. Phys. Vol.43 (2004) 739

