

行政院國家科學委員會研究計畫成果報告

計畫題目: 極平整之超薄氧化層

計畫編號: NSC 88-2215-E-009-033

執行期限: 87年8月1日至 88年7月31日

主持人: 蔡中教授 執行單位: 交通大學電子工程系 學生: 石坤桓

一、中文摘要

元件的scaling down已經是未來不可避免的趨勢之一。隨著MOSFET中oxide厚度的減小，若想要防止direct-tunneling current density增加，那麼oxide的完整性就很重要。

我們使用一種新的製程技術，用來去除原生氧化層以成長出高品質的gate oxide。利用此技術，我們可以成長出均勻度高、平滑性佳、漏電流低的thin gate oxide。

關鍵詞: 均勻度、平滑性、漏電流

Abstract We have studied the inversion layer mobility of nMOSFET's with thin-gate oxide of 20 to 70 Å. Direct relationship of electron mobility to oxide/channel interface roughness was obtained from measured mobility of MOSFET's and high-resolution TEM. By using a low-pressure oxidation process with native oxide removed *in situ* prior to oxidation, atomically smooth interface of oxide/channel was observed by high-resolution TEM for oxide thicknesses of 11 and 38 Å. The roughness increased to one to two monolayers of Si in a 55-Å oxide. Significant mobility improvement was obtained from these oxides with smoother interface than that from conventional furnace oxidation. Mobility reduction with decreasing oxide thickness was observed in the 20 and 35-Å oxide, with the same atomically smooth oxide/channel interface. This may be due to the remote Coulomb scattering from gate electrode or the gate field variation from poly-gate/oxide interface roughness.

keywords: TEM, monolayer and atomically smooth

Introduction

By continuously scaling the thickness of gate oxide, both the current drive capability and the transconductance of MOSFET's increase [1]-[5]. Current drive more than 1.0 mA/cm² and transconductance over 1000 mS/mm are reported for deep submicrometer MOSFET's with a 10Å direct tunneling oxide [4]. This performance can ensure the transistors to operate at low battery voltage for portable wireless communication [4], [6]. However, the most important issues for MOSFET made by ultrathin gate oxide are thickness uniformity and interface smoothness. The interface roughness can strongly affect the carrier transport that can be characterized by measuring the electron mobility in the inversion layer of MOSFET's [7]-[9]. Furthermore, electron mobility is an important parameter for device modeling and design, and the speed of MOSFET is dependent on the mobility at the low electric field near source. In this letter, we have measured the electron mobility with different interface roughness. Very smooth interface of oxide/Si is achieved by desorbing the native oxide *in situ* in a low-pressure oxidation system before thermal oxidation. Atomically smooth interface between oxide and Si is observed by high-resolution TEM for oxide thicknesses of 11 and 38Å. The electron mobility dependence on oxide thickness of 20 to 70Å was also measured, and a mobility reduction is observed in thinner oxide.

Manuscript received January 17, 1997; revised May 29, 1997.

A. Chin, T. Chang, R. H. Kao, S. C. Lin, and C. Tsai are with the Institute of Electronics Engineering, National Chiao Tung University, Hsinchu 300, Taiwan, R.O.C.

W. J. Chen is with the Department of Mechanical Materials Engineering, National Yun-Lin Polytechnic Institute, Huwei, Taiwan, R.O.C.

J. C.-M. Huang is with ERSO, ITRI, Hsinchu, Taiwan, R.O.C.

二、 實驗方法

To reduce the interface roughness of oxide/Si, we have designed a low-pressure oxidation system that can desorb native oxide *in situ* before oxidation. A leak-tight reactor design and a high flow rate of hydrogen are used to avoid further growth of native oxide after wafer loading. Similar leak-tight technique and native oxide desorbing process have been used for low-pressure chemical vapor deposition (LPCVD) to grow high quality Si epitaxy at 550C [10]. P-type 4-in [100] Si wafers with typical resistivity of 10 Ω -cm are used in this study, and the oxide is grown at 900C using N₂O under a reduced pressure of 4.5 torr. The advantages of the low pressure oxidation are slow oxidation rate for precise thickness control and good thickness uniformity due to increased N₂O migration length. Ultrathin oxide of 11Å [Fig. 1(b)] can be reproduced and thickness variation less than 1Å is obtained across 4-in wafer for a 20Å oxide. More detailed growth of ultrathin oxide and process flow will be published elsewhere. High-resolution TEM and 100-lbm wide MOSFET are used to characterize the interface roughness and mobility behavior. To avoid the drain voltage (V_d) bias dependence, the electron mobility was determined from the drain conductance and the gate-channel capacitance at a low V_d of 50 mV [8]. The effective normal field can be expressed as

$$E_{eff} = (Q_{inv}/2 + Q_B) / \epsilon_{si}$$

where Q_{inv} is the inversion layer charge, Q_B is the bulk depletion-layer charge and ϵ_{si} is the permittivity of Si.

三、 結論與討論

Fig. 1(a) and (b) show the atomic image of a 38- and 11Å oxide grown in the low-pressure oxidation system with native oxide removed *in situ*. Atomically smooth interface of oxide/Si can be observed in Fig. 1, and only one atomic layer of Si, just beneath the oxide, is disturbed from its original crystal structure. It is noticed that there are only two Si-atomic layers oxidized to form the ultrathin oxide of 11Å.

We have also grown oxide from conventional furnace for comparison. Fig. 2(a) presents a ~55Å oxide grown by a conventional furnace, where both interface waving and roughness up to four to five layers of Si are observed. In contrast, as shown in Fig. 2(b), the interface of oxide/Si is much smoother for the same oxide thickness grown in our system with native oxide removed *in situ*. The local thickness variation up to one to two Si-atomic layers observed in Fig. 2(b) may be due to the increased thermal stress of the thicker oxide than that shown in Fig. 1.

Based on the smooth oxide/Si interface, we have therefore studied the electron mobility improvement due to reduced interface roughness. Wafers with low concentration of impurity doping are used to reduce the ionized impurity scattering in the inversion layer, which dominates the low-field mobility. Fig. 3 shows the measured mobilities with 70-Å oxide from conventional furnace oxidation and from low-pressure oxidation with native oxide removed *in situ*, respectively. The measured electron mobility from conventional furnace oxidation follows the universal mobility-field dependence reported in previous papers [7]-[9]; however, significant improvement of mobility is measured from our low-pressure oxidation with a clean surface free of native oxide. It is important to note that both wafers from conventional furnace oxide and low-pressure oxide were fabricated at the same time, and they have the same thickness of poly-gate and doping concentrations at source and drain. Therefore the higher electron mobility measured from low-pressure oxide is not process-related. The improved mobility is due to the reduced interface roughness scattering [7]-[9]; however, significant improvement of mobility is measured from our low-pressure oxidation with a clean surface free of native oxide. It is important to note that both wafers from conventional furnace oxide and low-pressure oxide were fabricated at the same time, and they have the same thickness of poly-gate and doping concentrations at source and drain. Therefore the higher electron mobility measured from low-pressure oxide is not process-related. The improved mobility is due to the reduced interface roughness scattering [7]-[9]; however, significant improvement of mobility is measured from our low-pressure oxidation with a clean surface free of native oxide. It is important to note that both wafers from conventional furnace oxide and low-pressure oxide were fabricated at the same time, and they have the same thickness of poly-gate and doping concentrations at source and drain. Therefore the higher electron mobility measured from low-pressure oxide is not process-related. The improved mobility is due to the reduced interface roughness scattering [7].

We have also measured the effective mobility as a function of gate oxide thickness, for oxide thicknesses of 20-, 35-, and 70-Å, respectively. As shown in Fig. 4, a mobility decrease in the low gate-field region is observed for the 20 and 35-Å oxide as compared to the 70-Å one. It is well known that the total electron mobility is governed by individual contributions from phonons, channel doping impurities and interface roughness, where the Coulomb scattering from doping impurities dominates the low-field mobility and the interface roughness dominates the high-field mobility [8], [9]. Therefore interface roughness scattering limits the mobility of modern deep submicrometer devices even though the channel doping is higher than that used in this work [7]. Because the same resistivity of substrates are used, the reduction in low-field mobility

is not due to the different concentrations of impurities. Although the decay of mobility as decreasing oxide thickness is generally explained by the over-estimation of inversion carrier concentrations in thin oxides [2], [11], other possibilities such as Coulomb scattering from remote charge at poly-gate or the poly-gate/oxide interface roughness may also be responsible to the mobility degradation. The remote charge scattering is due to poly-gate depletion [12] and is well known in III-V modulation-doped FET and was also predicted in [13] and [14]. The interface roughness of polygate and oxide, shown in Figs. 1 and 2, may also contribute local gate field variation [15], [16] and provide additional scattering mechanism to reduce the mobility at thin oxide.

Conclusion

In conclusion, we have shown direct relationship of electron mobility to oxide/channel interface roughness. Significant mobility improvement is obtained from a smoother interface with *in situ* removing native oxide prior to oxidation. Atomically flat interface of oxide/channel can be obtained by this method for oxide thickness in the range of 20-38 Å. Mobility reduction with decreasing oxide thickness was observed in the 20- and 35-Å oxide, with the same atomically smooth oxide/channel interface. This may be due to the remote Coulomb scattering from gate electrode or the gate field variation from polygate/oxide interface roughness.

四、参考文献

- [1] J. Ahn, W. Ting, I. Chu, S. Lim, and D. L. Kwong, "High quality thin gate oxide prepared by annealing low-pressure chemical vapor deposited SiO₂ in N₂O," *Appl. Phys. Lett.*, vol. 59, no. 3, pp. 283-285, 1991.
- [2] C. I. Liu, E. J. Lloyd, Y. Ma, M. Du, R. L. Opila, and S. J. Hillenius, "High performance 0.2- μ m CMOS with 25 Å gate oxide grown on nitrogen implanted Si substrates," in *IEDM Tech. Dig.*, 1996, pp. 499-502.
- [3] C. Lin, A. Chou, K. Kumar, P. Chowdhury, and J. C. Lee, "Leakage current, reliability characteristics and boron penetration of ultra-thin (32-36 Å) O₂-oxide and N₂O/N₂O oxynitrides," in *IEDM Tech. Dig.*, 1996, pp. 331-334.
- [4] H. S. Momose, M. Ono, I. Yoshitomi, I. Ohguro, S. Nakamura, M. Saito, and H. Iwai, "1.5 nm direct-tunneling gate oxide Si MOSFET's," *IEEE Trans. Electron Devices*, vol. 43, pp. 1233-1241, Aug. 1996.
- [5] I. Matsumoto, S. Kakimoto, M. Nakano, H. Kotaki, S. Hayashida, K. Sugimoto, K. Adachi, S. Morishita, K. Uda, Y. Sato, M. Yamamoto, I. Ogura, and J. Takagi, "Direct tunneling N₂O gate oxynitrides for low-voltage operation of dual gate CMOSFET's," in *IEDM Tech. Dig.*, 1995, pp. 851-854.
- [6] R.-H. Yan, D. Monro, J. Weir, A. Mujtaba, and E. Westerwick, "Reducing operating voltage from 3, 2, to 1 volt and below-challenges and guidelines for possible solutions," in *IEDM Tech. Dig.*, 1995, pp. 55-58.
- [7] J. Hauser, "Extraction of experimental mobility data for MOS devices," *IEEE Trans. Electron Devices*, vol. 43, pp. 1981-1988, Nov. 1996.
- [8] S. C. Sun and J. D. Plummer, "Electron mobility in inversion and accumulation layers on thermally oxidized silicon surfaces," *IEEE Trans. Electron Devices*, vol. ED-27, pp. 1497-1508, Aug. 1980.
- [9] H. Shim, G. M. Yeric, A. F. Tasch, and C. M. Maziar, "Physically based model for effective mobility and local-field mobility of electrons in MOS inversion layers," *Solid-State Electron.*, vol. 34, no. 6, pp. 545-552, 1991.
- [10] A. Chin, B. C. Lim, and W. J. Chen, "High quality epitaxial Si grown by a simple low-pressure chemical vapor deposition at 550 degrees C," *Appl. Phys. Lett.*, vol. 69, no. 11, pp. 1617-1620, 1996.
- [11] M.-S. Liang, J. Y. Choi, P.-K. Ko, and C. Hu, "Inversion-layer capacitance and mobility of very thin gate-oxide MOSFET's," *IEEE Trans. Electron Devices*, vol. ED-33, pp. 409-413, Mar. 1986.
- [12] C. Hu, "Gate oxide scaling limit and projection," in *IEDM Tech. Dig.*, 1996, pp. 319-322.
- [13] A. A. Grimberg and M. S. Shur, "Effect of image charges on impurity scattering of two-dimensional electron gas in AlGaAs/GaAs," *J. Appl. Phys.*, vol. 58, no. 1, pp. 382-386, 1985.
- [14] F. Stern and W. E. Howard, "Properties of semiconductor surface inversion layers in the electric quantum limit," *Phys. Rev.*, vol. 163, no. 3, pp. 816-835, 1967.
- [15] M. Y. Hao and J. C. Lee, "Electrical characteristics of oxynitrides grown on textured single-crystal silicon," *Appl. Phys. Lett.*, vol. 60, pp. 445-447, 1992.
- [16] S. L. Wu, C. L. Lee, and I. F. Lei, "Tunnel oxide prepared by thermal oxidation of thin polysilicon film on silicon," *IEEE Trans. Electron Devices*, vol. 14, pp. 379-381, Aug. 1993.

Figure Captions:

Fig1. Cross-sectional TEM images (a) 38Å oxide and (b) 11Å oxide grown at 900C using N₂O under a low pressure of 4.5 Torr. Native oxide is desorbed *in situ* prior to oxidation.

Fig2. Cross-sectional TEM images of 55Å from (a) conventional oxidation at 800C and (b) from low-pressure oxidation with native oxide desorbed *in situ*.

Fig3. Electron mobilities of MOSFET's with 70Å oxide from conventional furnace oxidation and from low-pressure oxidation with native oxide desorbed *in situ*.

Fig4. Electron mobilities of MOSFET's as a function of effective field of (a) 70Å (b) 35Å and (c) 20Å oxide from low-pressure oxidation with native oxide desorbed *in situ*.

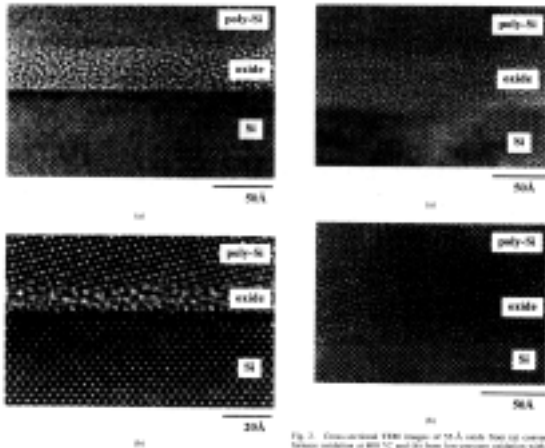


Fig. 1. Cross-sectional TEM images of (a) 70 Å oxide and (b) 35 Å oxide grown at 500 °C using N_2O as oxidant in low-pressure of 4.7 torr. Native oxide is desorbed *in situ* prior to oxidation.

Fig. 2. Cross-sectional TEM images of 20 Å oxide from (a) conventional furnace oxidation at 800 °C and (b) from low-pressure oxidation with native oxide desorbed *in situ*.

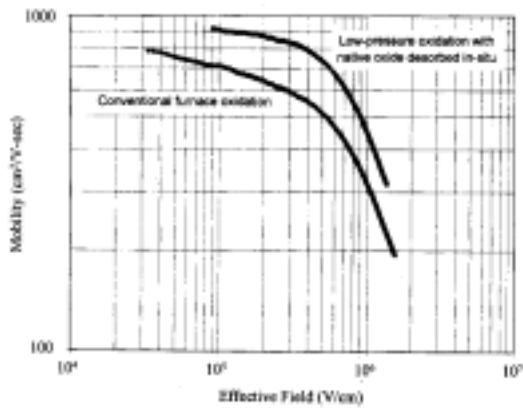


Fig. 3. Electron mobilities of MOSFET's with 70-Å oxide from conventional furnace oxidation and from low-pressure oxidation with native oxide desorbed *in situ*.

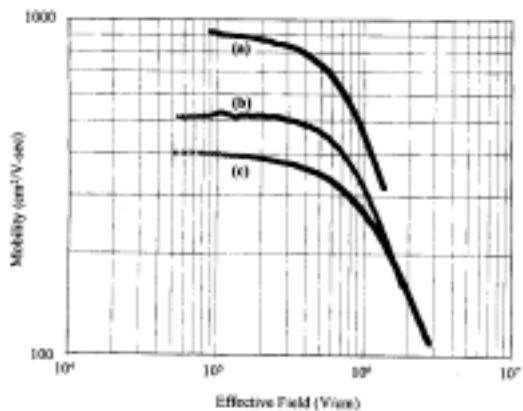


Fig. 4. Electron mobilities of MOSFET's as a function of effective field of (a) 70 Å, (b) 35 Å, and (c) 20 Å oxide from low-pressure oxidation with native oxide desorbed *in situ*.